Bond strength of porcelain to cobalt-chromium dental alloy fabricated by selective laser melting and casting methods

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ABSTRACT

This study aimed to compare the bond strength of porcelain to cobalt-chromium dental alloy fabricated by selective laser melting (SLM) and casting methods. Twelve rectangular bars measuring 25 x 3 x 0.5 mm were fabricated of cobalt-chromium alloy for each of the SLM and casting groups according to ISO9693:1999. Porcelain was applied at the center of each bar measuring 3 x 8 mm with 1 mm thickness. Three-point flexural bond strength test was performed to assess the bond strength of porcelain to alloy. Data were analyzed and compared between the SLM and casting groups via Independent sample t-test (alpha=0.05). Mode of failure was also determined by visual inspection of samples. The mean bond strength of porcelain to SLM alloy (35.26±1.22 MPa) was significantly higher than that to casting alloy (33.21±3.02 MPa) (P<0.05). Most failures in both groups were mixed and no sample showed adhesive failure. The SLM metal-ceramic system showed higher bond strength than the required threshold by ISO9693:1999. Compared to alloys fabricated by the casting method, the SLM method showed higher bond strength to porcelain. This relatively new technology is promising for dental application and can serve as a suitable alternative to the conventional casting method for the fabrication of metal-ceramic restorations.

KEY WORDS: SHEAR STRENGTH; METAL-CERAMIC; LASER MELTING

ARTICLE INFORMATION:

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Received 23rd June, 2017
Accepted after revision 24th Sep, 2017
BBRC Print ISSN: 0974-6455
Online ISSN: 2321-4007 CODEN: USA BBRCBA
Thomson Reuters ISI ESC and Crossref Indexed Journal
NAAS Journal Score 2017: 4.31 Cosmos IF: 4.006
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Online Contents Available at: http://www.bbrc.in/
DOI: 10.21786/bbrc/10.3/15

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INTRODUCTION

Metal-ceramic systems for dental restorations are available since the 1960s and the success of metal-ceramic restorations depends on the strength and uniformity of bond at the metal-porcelain interface, which is the most sensitive area in terms of crack formation (Bowers 1985) and (Drummond and et al 1984, Pavlović 2017 and Abduo 2017). Cracks mainly occur at the metal-ceramic interface or within the veneering porcelain (Daftary 1986) and (Zhukovsky 1996). Factors such as trauma, fatigue, occlusal loads and incompatibility between the mechanical properties of metal and porcelain may result in porcelain fracture especially of cohesive type (Pameijer 1996 and Kelsey 2000, Ren 2016 and Kaleli 2017). Metal-ceramic frameworks are conventionally fabricated using the lost wax casting (Anusavice 2003) and (Kaleli 2017).

However, problems associated with casting of base metal alloys such as high melting point, high oxidation potential during casting, time consuming nature of casting and finishing of these alloys, limit their application. Thus, application of a technique to eliminate the casting process may enhance the use of these alloys (Bhaskaran 2013) and (Akova 2008).

The selective laser melting (SLM) technology, also known as the three-dimensional (3D) printing, was introduced for the fabrication of metal copings in metal-ceramic crowns. This technique does not have many of the limitations of the conventional waxing technique, so that the restorations are fabricated by incremental application of 0.02 μ layers (Xin 2012). This technique works based on the computer aided design (CAD) data obtained from the framework design, and uses high temperature laser beams for selective melting of metal framework, which is in the form of powder. The metal framework is formed by incremental addition of these thin layers (Akova 2008).

The success of SLM is related to its ability in fabrication of metal components with complex geometrical shapes from a 3D CAD model. Moreover, this technique can produce components with mechanical properties comparable or superior to those of conventionally fabricated components (Örtorp 2011 and (Quante 2008). Thus, it can be an alternative to the conventional process. Only a few studies have assessed the bond strength of porcelain to alloys fabricated by the SLM technique. Some of these studies have reported superior mechanical properties for copings fabricated by this method and higher bond strength of porcelain compared to those fabricated by the conventional casting method, while some others found no significant difference in bond strength of porcelain between the two groups (Liu YH 2010) and (Wu 2014) and (Ren 2016).

Considering the gap of information and the controversies in this regard, this study aimed to assess the bond strength of cobalt-chromium (CoCr) alloy fabricated by the SLM technique and the conventional wax burnout and casting method.

MATERIALS AND METHODS

This in vitro, experimental study was conducted on 24 samples (12 fabricated by conventional casting and 12 fabricated by the SLM technique). In the casting group, 12 rectangular bars were fabricated measuring 25 mm ± 1 mm × 3 mm ± 0.1 mm × 0.5 mm ± 0.05 mm according to the ISO9693-1 (The International Organization for Standardization 2012) using green wax sheets (Berg, Karl Berg GmbH, Engen, Germany). They were sprayed and flaked using phosphate-bonded investment material. After wax burnout at 650°C, melted CoCr was poured into the space created by wax burnout using a standard broken-arm centrifugal casting machine (Kerr, Orange, CA, USA). Metal sheets were cleaned from investment material and the sprues were cut. Dimensions were adjusted and the thickness of sheets was ensured by a caliper and gauge.

In the SLM group, 12 samples were fabricated with the same dimensions as those in the casting group using EOSINT M270 machine (Manufacturing system; Bego, Bremen, Germany). This device was equipped with ytterbium laser fiber with less than 200 W laser power, scanning speed of 5-10 mm/second, wavelength of 1060-1100 nm, beam diameter of 100-500 μm and powder feed rate of 5-7 g/minute. Using data obtained from a CAD file with STL format, the device fused metal particles and after scanning each cross-section, the alloy thickness decreased by one layer and a new layer of alloy was applied on the upper surface. This process was repeated until the entire restoration was formed. Table 1 shows the composition of CoCr alloy used for the casting (WBC 9581TM, Bego, Germany) and SLM (EOS Cobalt Chrome SP2; EOSGmbH, Germany) techniques.

The samples in both groups were then subjected to air-borne particle abrasion for five seconds (with 110-250 μ aluminum oxide particles at 3-4 MPa pressure for the SLM alloy and 120 μ aluminum oxide particles at 0.35 MPa pressure for the casting alloy) to provide surface roughness similar to the standard process practiced in the clinical setting. The samples were then cleaned with vapor spray for six seconds. Pre-oxidation was then performed according to the manufacturer’s instructions at 950-980°C for five minutes for SLM and 980°C for eight minutes for the casting alloy. Air-borne particle abrasion was then performed again and before porcelain application, the samples were cleaned in an ultrasonic...
Table 1. Composition of CoCr alloy used for the casting and SLM techniques according to the manufacturers (wt%)

<table>
<thead>
<tr>
<th>Group</th>
<th>Co</th>
<th>Cr</th>
<th>Mo</th>
<th>W</th>
<th>Si</th>
<th>Nb</th>
<th>V</th>
<th>Fe</th>
<th>Mn</th>
<th>Ce</th>
</tr>
</thead>
<tbody>
<tr>
<td>SLM alloy</td>
<td>63.9</td>
<td>25.3</td>
<td>5.2</td>
<td>5.5</td>
<td>1</td>
<td>-</td>
<td>-</td>
<td>Max 0.5</td>
<td>Max 0.1</td>
<td>-</td>
</tr>
<tr>
<td>Cast alloy</td>
<td>61</td>
<td>26.6</td>
<td>5</td>
<td>&lt;2</td>
<td>-</td>
<td>-</td>
<td>&lt;2</td>
<td>-</td>
<td>&lt;2</td>
<td>-</td>
</tr>
</tbody>
</table>

Co, cobalt; Cr, chromium; Mo, molybdenum; W, tungsten; Si, silicon; Nb, niobium; V, vanadium; Fe, iron; Mn, manganese; Ce, Cerium; SLM, selective laser melting; max, maximum.

bath containing 95% ethanol (Elmasonic, Elma Hans GmbH & Co, Koln, Germany) for five minutes.

Next, equal layers of opaque and body porcelain (VMK 95, Vita, Bad Säckingen, Germany) with 8 mm length, 3 mm width and 1 mm thickness (0.2 mm opaque porcelain and 0.8 mm body porcelain) were applied and baked according to the manufacturer’s instructions (Figures 1 and 2). For the purpose of standardization of porcelain thickness, a silicon index was used for all samples (Figure 3). To minimize the effect of confounders, all procedures were performed by one operator.

A three-point flexural strength test was performed according to the ISO9693-1 standard using a universal testing machine (Z020, Zwick Roell, Germany). Surface of the sample with porcelain over it was placed downward and the two ends of each rod were placed on the supporting fulcrums with 0.9 mm diameter and 20 mm distance from each other. Load was applied to the center of each sample by a crosshead with 0.9 mm diameter at a speed of 1 mm/minute (Figure 4). Load application was continued until failure (abrupt drop in the stress-strain curve). The load at failure was recorded and reported by the computer. Bond strength in megapascals (MPa) was calculated using the formula \( A = k \times F \) (N/mm²) where \( k \)
is a constant related to thickness and modulus of elasticity of the alloy and $F$ is the maximum load in Newtons causing debonding.

After conduction of flexural test, the samples were visually inspected to determine the mode of failure. Mode of failure was divided into three groups of adhesive (at the metal-porcelain interface), cohesive (completely within the porcelain) and mixed (a combination of adhesive and cohesive types).

Finally, data were analyzed by Independent samples t-test.

RESULTS AND DISCUSSION

Table 2 shows the mean and standard deviation of bond strength in the two groups.

Independent samples t-test was used to compare the mean values between the two groups, which showed that the mean bond strength of porcelain was significantly higher to alloy in SLM group compared to the casting group ($p=0.04$).

Inspection of the surface of samples in the SLM group before porcelain applying showed that the surface of bars in the SLM group was dark gray without metal shine. Figure 5 shows photography of the surface morphology of metal matrix of the one sample of each group after porcelain debonding. Visual inspection of each sample during flexural strength testing showed that debonding cracks between metal and ceramic occurred at one end of the ceramic layer and not at the center.

In terms of mode of failure, in the SLM group, most failures were mixed and no adhesive failure was noted. In the casting group, all failures were mixed (Table 3).

DISCUSSION

Failure of the metal-porcelain bond can occur as the result of a combination of factors such as different coefficients of thermal expansion of metal and ceramic, presence of micro-cracks within the porcelain and occlusal loads or trauma. Loss of porcelain to metal bond is the second most common cause of replacement of metal-ceramic restorations [18]. The reason for use of SLM technology in this study was the recent introduction of this technique and its numerous advantages. The SLM technique enables the fabrication of restorations with a more uniform quality and standardization of restoration shaping process with lower cost, requiring less human force, saving time due to elimination of many procedural steps (compared to the conventional method) and elimination of human errors. Moreover, CAD, compared to manual waxing, results in more accurate control of the porcelain thickness and decreased risk of fracture. SLM scanners have disadvantages as well including limitation of scanning systems in terms of resolution and creating rounder margins as well as creation of cloudy spots caused by scanning, which result in internal mismatch (Walton 1986) and (Huang 2015) and (Li 2014). High cost of laser device used in this technique is another disadvantage of this method. Also, the CoCr alloy...
fabricated by the SLM method has not shown any cytotoxic potential and evidence shows that it has higher corrosion resistance and lower release of cobalt ion compared to the casting group (Xin 2012).

Among the suggested tests for assessment of bond strength, Schwickerath crack initiation test or three-point flexural strength test was first suggested by Lenz et al, and is also recommended by the ISO9693:1999(E) for determination of debonding strength of metal-ceramic systems (Lenz 1995). Although several mechanical tests such as flexural, torsional, shear, tensile or a combination of flexural and shear tests can be used for assessment of the bond strength at the metal-ceramic interface, the validity of the bond strength assessment tests is still questionable (Kontonasaki 2008).

The three-point flexural strength test reveals the modulus of elasticity, flexural strain, flexural stress and flexural stress-strain response of a material. The main advantage of this test is easy sample preparation and conduction. However, it has drawbacks as well. One drawback is that the test result is affected by the geometry of samples, load application and speed of strain application (Kontonasaki 2008). For this reason, in the current study, we tried our best to control for all the confounding factors that had the potential to affect the results such as speed of load application (crosshead speed), distance between the two fulcrums and dimensions of the samples. All samples were fabricated with the same dimensions. Also, a custom-made silicon index was used for control of the length and thickness of ceramic layer when applying the porcelain.

In order to acceptably compare the bond strength of ceramic to metal between restorations fabricated by two different techniques, the materials used should be the same in terms of behavior with regard to the formation of oxide layer, coefficient of thermal expansion, etc., as much as possible (Kontonasaki 2008) and (Zinelis 2003). Therefore, in the current study, alloys with the most similar composition were chosen for SLM and casting groups in order to minimize the effect of elements in the composition of alloys and assess the pure effect of fabrication method on the results.

The bond between the porcelain and metal can be achieved by three mechanisms of van der Waals forces, mechanical retention and chemical bond; chemical bond is the dominant factor responsible for metal-porcelain bond. Chemical bond is influenced by the composition of elements in the alloy and formation of oxide layer on the metal surface. Due to the significance of the thickness of oxide layer in a successful bond, the pre-oxidation phase of alloy was performed for both groups in our study prior to the addition of porcelain according to the manufacturer’s instructions (Bagby 1990) and (Jochen 1986).

In this study, the SLM group showed higher porcelain bond strength compared to the casting group, which was in line with the results of Liu et al [14]. One possible reason for higher bond strength of the SLM group was the surface properties of this alloy. The SLM system works based on incremental application of melted powders. The surfaces of fabricated components by the SLM technique often have sticky powder. In other words, relatively melted powders are added to the surface of components and make them rougher (Zhang 2014). Due to the inherent roughness of the surface of SLM samples and according to the results of a study by Zhang (Zhang 2014), it may be stated that these surfaces can increase the contact area of framework and porcelain. At the same time, surface porosities can provide strong micromechanical retention for porcelain and enhance the bond strength (Zhang 2014).

Another possible explanation is penetration of porcelain into the gaps created between layers as well as the balling phenomenon during incremental fabrication by the laser sintering process, which increase the contact area and bond strength by creation of undercuts (Eunjung 2014) and (Gu D 2007).

The fact that all tested samples in our study yielded higher bond strength than the minimum strength required by ISO9693:1999(E), which is 25 MPa, shows that the alloy fabricated by the SLM technique can not only provide an acceptable bond for clinical application, it even yields higher bond strength than the casting method. Akova and Serra-Prat in separate studies showed that despite higher bond strength of the casting group, the difference between the casting and SLM techniques did not reach statistical significance (Serra-Prat 2014). Controversy between their results and ours may be attributed to the different form of samples (cylindrical) and assessment of shear bond strength instead of flexural bond strength.

Xiang and Ren found no significant difference in bond strength of the SLM and casting groups; although the values obtained in the SLM group were higher in both studies (Xiang 2012). Difference between their results and ours may be due to difference in the load application speed for flexural test, difference in laser system and composition of alloys used and size of sandblasting particles.

Inspection of the surface of samples in the SLM group before porcelain applying revealed that the surface of bars in the SLM group was dark gray and had no metal shine in our study. The reason may be that the metal powder particles are fused on the surface of components fabricated by the SLM method; thus, the samples show a color similar to that of metal powder particles.

Visual observation of each sample during the flexural strength testing revealed that the debonding cracks at
the metal-ceramic interface occurred at one end of the ceramic layer and not at the center of it, which was in agreement with ISO9693:1999(E) standards. Evaluation of surface morphology of metal matrix after porcelain debonding showed that the color was different at the center and at the ends of the samples. This indicated that the ceramic layer was still present on the entire surface of samples. Moreover, mode of metal-ceramic debonding may indicate the fracture energy that degrades the metal-ceramic bond. The higher the amount of porcelain body remaining on the metal surface, the higher the adhesion of porcelain body to metal, the higher the fracture energy and the lower the risk of porcelain fracture would be in the clinical setting (Wagner 1993) and (Lavine 1966). Five samples in the SLM group showed cohesive failure while in the casting group, all fractures were mixed. The reason is probably the higher bond strength of the SLM group. Since after debonding high amounts of porcelain remained on the alloy surface, it may be concluded that a strong bond was created between dental porcelain and SLM alloy at the interface. Thus, this technique can increase the success rate of porcelain fused to metal restorations.

Since this study had an in vitro design, the SLM alloy should also be evaluated in the oral cavity. Also, it is recommended to assess the marginal fit of copings fabricated by this technique. The bonding interface should be evaluated using scanning electron microscope and energy-dispersive X-ray spectroscopy in future studies. The microstructure of SLM alloy must also be analyzed under a metallographic microscope.

CONCLUSION

The SLM alloy system not only provides a clinically acceptable bond strength, the bond strength of restorations fabricated by this method to porcelain was even higher than that of restorations fabricated by the conventional casting method. Thus, this relatively new technology can serve as an alternative to the conventional casting method for the fabrication of metal-ceramic restorations.

ACKNOWLEDGEMENTS

The work described in this paper has been retrieved from Dr. Parisa Alirezaei’s thesis. The authors would like to thank Dr. Marzieh Mahmoodi for useful statistical consultations.

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